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& INTERFACES

Effects of Substrate on Piezoelectricity of Electrospun Poly(vinylidene fluoride)-Nanofiber-Based Energy Generators

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ABSTRACT: We report the effects of various substrates and substrate thicknesses on electrospun poly(vinylidene fluoride) (PVDF)-nanofiber-based energy harvesters. The electrospun PVDF nanofibers showed an average diameter of 84.6 ± 23.5 nm. A high relative β-phase fraction (85.2%) was achieved by applying high voltage during electrospinning. The prepared PVDF nanofibers thus generated considerable piezoelectric potential in accordance with the sound-driven mechanical vibrations of the substrates. Slide glass, poly(ethylene terephthalate), poly(ethylene naphthalate), and paper substrates were used to investigate the effects of the intrinsic and extrinsic substrate properties on the piezoelectricity of the energy harvesters. The thinnest paper substrate (66 μ m) with a moderate Young's modulus showed the highest voltage output (0.4885 V). We used high-performance 76, 66, and 33 μm thick papers to determine the effect of paper thickness on the output voltage. The thinnest paper substrate resulted in the highest voltage output (0.7781 V), and the numerical analyses of the sound-driven mechanical deformation strongly support the hypothesis that substrate thickness has a considerable effect on piezoelectric performance.

KEYWORDS: poly(vinylidene fluoride), electrospun nanofiber, piezoelectric nanogenerator, paper

1. INTRODUCTION

Rechargeable power and renewable energy sources have recently attracted significant attention because of deepening concerns about the fossil fuel depletion and pollution caused by excessive energy consumption. Various forms of reusable energy—such as electrical power sources (i.e., energy harvest ers)—are promising solutions to these problems. Using nanomaterials to convert various energies (such as solar, wind, thermal, tidal, etc.) into electrical energy has been researched extensively.1−⁴ Mechanical vibration is a renewable energy more abundant than many other sources since all movements from ato[ms](#page-6-0) to the earth produce mechanical vibration. Sound waves are mechanical vibrations propagating through air and are both the most abundant and most wasted energy. For example, train, airplane, and car sounds produce mechanical vibrations in the order of tens to hundreds of decibels.⁵

Piezoelectric materials are materials that accumulate electrical charge i[n](#page-6-0) response to applied mechanical stress. They can be classified as natural (e.g., berlinite, quartz, topaz, 8° etc.) or synthetic (e.g., $BaTiO₃⁹$ PbTiO₃,¹⁰ lead zirconate titanate (PZT) ,¹¹ ZnO,¹² poly(vinylidene [fl](#page-6-0)uoride) (PVD[F\)](#page-6-0),¹³ etc.). BaTiO₃, PZT, ZnO, [a](#page-6-0)nd PVD[F](#page-6-0) are the most widely investi[gat](#page-6-0)ed pi[ezo](#page-6-0)electric materials owing to their [su](#page-6-0)perior piezoelectric performance.¹⁴ PVDF was recently used as the core material in a wearable/implantable piezoelectric transducer

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Figure 1. Schematics of the (a) electrospinning setup and (b) nanogenerator. (c) SEM image of electrospun PVDF nanofiber morphology. (d) FT-IR spectrum showing peaks associated with the α and β phases.

because it is malleable,^{13,15} and it is lightweight,¹⁶ flexible,¹⁷ stretchable,¹⁷ and biocompatible.¹⁸ The mechanical-stressinduced microstructur[al de](#page-6-0)formation of piezoel[ect](#page-6-0)ric tra[ns](#page-6-0)ducers pro[du](#page-6-0)ces dipole moments [tha](#page-6-0)t generate differences in potential at both transducer ends. Kawai first demonstrated the tensile piezoelectricity in stretched and poled PVDF films in 1969.¹⁹ Uniaxially oriented PVDF exhibits relatively weak electromechanical coupling and shows a piezoelectric coefficien[t](#page-6-0) of -33 pC·N⁻¹, significantly poorer than inorganic perovskite piezoelectrics.^{20,21} This limitation apparently stems from randomly oriented H−C−F dipoles, as untreated PVDF can contain α and β cry[stallin](#page-6-0)e phases. The nonpolar α phase, consisting of randomly oriented dipoles, forms most readily when PVDF is cooled above the glass-transition temperature (T_{σ}) , while the well-oriented dipoles form the piezoelectric β phase.²² Thus, post-treating PVDF to convert the α into the β phase improves piezoelectric performance. However, it is also costly [an](#page-6-0)d time-consuming, so this step is usually omitted when piezoelectric nanogenerators are fabricated in an industrial setting. Researchers have recently observed in situ the relationship between the mechanical stretching and electrical poling of piezoelectric nanogenerators and used direct-written (or near-field electrospun) PVDF nanofibers to transform the α phase into the polar $\hat{\beta}$ phase at a high voltage.^{23−25} Moreover, Nasir et al. systematically investigated the microstructural phases of electrospun PVDF nanofibers,²⁶ w[hich m](#page-6-0)ay be the optimal material for improving the performance of piezoelectric nanogenerators. Nanogenerator substrate[s c](#page-6-0)an greatly affect the performance of nanomaterial-based piezoelectric transducers.

Thin-film structured substrates can effectively convert sound waves into mechanical vibration and eventually transfer the mechanical vibration to the electrospun PVDF nanofibers. Therefore, combining sound waves and piezoelectric transducers may facilitate the use of mechanical vibration as electrical energy because sound wave sources are abundant and piezoelectric materials exhibit effective energy-conversion mechanisms. To our knowledge, the effects of various substrates and their thicknesses on PVDF-nanofiber-based piezoelectric nanogenerators have not been investigated to date. Therefore, we systematically analyzed the effects of the Young's moduli and thicknesses of various substrates on PVDF nanofiber piezoelectricity and identified the optimal substrate for application in energy harvesters.

2. EXPERIMENTAL SECTION

Electrospinning PVDF Nanofibers. The PVDF solution was produced from PVDF powder $(M_{w} = 534\,000\, \text{g/mol})$ (Sigma-Aldrich), N,N-dimethylformamide (DMF) (Sigma-Aldrich), acetone (Sigma-Aldrich), and ZonylUR (DuPont), which is an anionic phosphate fluorosurfactant that was provided without solvents. All reagents were used as purchased. The solution consisted of 16, 80, and 4 wt % PVDF, solvent (DMF:acetone = 4:6), and fluorosurfactant, respectively. The acetone and fluorosurfactant were used to improve the evaporation and reduce the surface tension of the solution, 27 which was homogenized at 100 °C for 1 day by stirring. Slide glasses (1125 μ μ μ m thick), poly(ethylene terephthalate) (PET) film (255 μ m thick), poly(ethylene naphthalate) (PEN) (105 μ m thick), and 33, 66, and 76 μ m thick papers whose densities were 25, 52, and 80 g/m² (OJI Specialty, View Corona), respectively, were used as the substrates. The

PVDF solution was electrospun at 0.5 mL/h onto the various substrates rotating at 800 rpm. The tip-to-collector distance was 15 cm, and the applied voltage was 15 kV. The electrospun nanofibers were collected onto two grounded aluminum electrodes on the substrates, which were separated by a 2.5 cm gap.

Characterization. The morphologies of the electrospun nanofibers were observed using scanning electron microscopy (SEM, S-4700). The sound waves were generated using a speaker and a function generator (Tektronix, CFG280), and the intensity of the generated sound was measured using a decibel meter. The effects of the sound waves on nanogenerator performance were assessed using an oscilloscope (Agilent, MS07054A). Fourier-transform infrared spectrometry (FT-IR, Jasco) was used in the 400–2000 cm^{-1} range to generate IR spectra for the electrospun fibers.

Simulation. The effects of substrate thickness on nanogenerator performance were simulated numerically. A nanogenerator on a paper substrate and a single straight PVDF filament laid across the two Al electrodes were assumed for simplicity.²⁸ The boundary-element method was used to solve nonsingular boundary integral equations to analyze piezoelectric performance on the [bas](#page-7-0)is of the free vibration of
the paper substrate.²⁹ A simply supported, 25 mm long, 31 mm wide rectangular plate was assumed to evaluate the vibration. The substrate width included the [wid](#page-7-0)ths of the two electrodes. The paper parameters were as follows: density =1.5 $g/cm³$, Young's modulus = 25 GPa, and Poisson's ratio = 0.3. The PVDF fiber parameters were as follows: fiber diameter = 5 μ m, fiber length = 25 mm, piezoelectric constants of d_{31} = 20 pC·N⁻¹, d_{32} = 3 pC·N⁻¹, d_{33} = -33 pC·N⁻¹, Young's modulus = 2.5 GPa, Poisson's ratio = 0.348, and relative dielectric constant = 8. The Al foil electrode parameters were as follows: width = 3 mm, length = 25 mm, thickness = 60 μ m, density = 27 g/cm³, Young's modulus = 70 GPa, and Poisson's ratio = 0.35. The MATLAB software was used for the numerical analyses.

3. RESULTS AND DISCUSSION

A schematic of the electrospinning setup is shown in Figure 1(a). The PVDF nanofibers were electrospun directly onto a nanogenerator device consisting of two Al foil electrodes on an [in](#page-1-0)sulating substrate. Conductive carbon tape was used to fix the device to the substrate. The collected nanofibers initially aligned because charge neutralization straightened them $30,31$ and then randomly oriented after several nanofiber layers had been stacked because the previously collected nanofi[bers](#page-7-0) inhibited charge neutralization and hence prevented the electrode from continuing to act as a countercharge reservoir. Thus, it is reasonable to assume that the PVDF nanofibers were randomly oriented on the substrate. The collection drum rotation affected the nanofiber orientation only negligibly because the collection drum linear rotation speed, on the order of several meters/second, was considerably slower than the nanofiber spinning speed (e.g., several kilometers/second).³² A schematic of the PVDF-nanofiber-based energy harvester is shown in Figure 1(b). The PVDF nanofibers were electro[spu](#page-7-0)n between two Al foil electrodes on the substrate, which vibrated when external s[ou](#page-1-0)nd waves induced voltage output through PVDF nanofiber dipole moments. Figure 1(c) shows an SEM image of PVDF nanofibers. The average nanofiber diameter $(84.6 \pm 23.5 \text{ nm})$ was less than that in p[re](#page-1-0)vious work.²⁶ The FT-IR spectra of the nanofibers are shown in Figure 1(d). Intense [p](#page-6-0)eaks associated with the PVDF nanofiber β phase were at ~470, 510, and 840 cm⁻¹, while weak peaks asso[cia](#page-1-0)ted with the α phase were evident at ~530, 610, and 763 cm^{-1.26} . The peaks at \sim 763 cm⁻¹ (related to CF₂ bending and skeletal bendin[g\)](#page-6-0) and 840 cm⁻¹ (related to CF_2 symmetric stretching) are characteristic of the α and β phases, respectively.^{33–36} The relative β -phase fraction can be determined using the characteristic peak wavenumbers and the following [equati](#page-7-0)on

Figure 2. Frequency optimized to evaluate electrospun-PVDFnanofiber-based energy harvester: (a) voltage-cycling-output profiles for various sound frequencies and (b) maximum voltage plotted as a function of sound frequency. The substrate was 100 μ m thick PET film.

$$
F(\beta) = \frac{X_{\beta}}{X_{\alpha} + X_{\beta}} = \frac{A_{\beta}}{\frac{K_{\beta}}{K_{\alpha}} A_{\alpha} + A_{\beta}}
$$
\n(1)

 X_{α} and X_{β} represent the degree of crystallinity of each phase; A_{α} and A_β represent the absorbencies of the α and β phases; and K_{α} and K_{β} are the absorption coefficients at each wavenumber and are 6.1×10^4 and 7.7×10^4 cm²/mol, respectively.³⁵ The resulting relative $β$ -phase fraction (85.2%) strongly suggests high-voltage-induced poling;³⁷ thus, the electrospun [PV](#page-7-0)DF nanofibers were expected to be adequate for fabrication of energy harvesters.

The electrical energy converted from the sound waves was displayed as an alternating current voltage on the oscilloscope (AC input mode) because sound-wave frequencies are sinusoidal. Sound is a regular mechanical vibration that travels as a waveform through matter. Longitudinal sound waves (i.e., compression waves) transmitted through ambient air are composed of waves whose alternating pressure deviates from the equilibrium pressure, causing local regions of compression and rarefaction.³⁸ Flexible substrates vibrate mechanically when struck by sound waves. Transferring vibrations directly to randomly align[ed](#page-7-0) PVDF fibers causes them to vibrate in turn. The sound-wave frequency was optimized to produce more significant results before evaluation of the nanogenerator

Figure 3. Sound-wave-driven voltage output according to substrate. (a) Profiles of overall voltage cycling output, (b) maximum voltage output for each substrate, (c) relationship between Young's modulus and maximum output voltage, (d) output-voltage-cycling profiles for PET (100 μ m) and PEN (105 μ m) films, and (e) relationship between substrate thickness and maximum output voltage. Substrate Young's moduli from ^aPaul et al.,³⁷ $^{\rm b}$ Karger-Kocsis et al., 38 'Sidler et al., 39 and $^{\rm d}$ Yokoyama et al. 40 are provided as reference values.

performance. Figu[re](#page-7-0) $2(a)$ [s](#page-7-0)hows the voltage cycling [p](#page-7-0)rofiles output for various sound frequencies. The dimensions are identical to those de[scr](#page-2-0)ibed in the Experimental Section, and the 100 μ m thick PET-film substrate was used. The sound power was 100 dB at 1 V input. Th[e lowest frequency \(50](#page-1-0) Hz) resulted in a repeatedly stable output voltage profile, but the

waveform was not sinusoidal in shape due to its sm[all](#page-7-0) amplitude. Moreover, the amplitude of the output voltage is less than 100 mV (Figure $2(b)$). At higher frequencies (300 and 500 Hz) the maximum voltages fluctuated, and the amplitudes were less than [20](#page-2-0)0 mV. The fluctuation of output voltage profiles at these higher frequencies may be attributed to

Figure 4. Relationship between substrate thickness and output voltage from PVDF nanofibers randomly aligned on paper versus PET film. Overall output voltages for (a) the paper substrate and (b) the PET substrate, as cycling continued. (c) Maximum output voltage according to substrate thickness.

a mismatch between the applied mechanical stress and the relaxation behavior of the PVDF nanofibers. Meanwhile, 100 Hz provided a highly stable oscillation and high amplitude (∼350 mV). Therefore, we retained 100 Hz as the input soundwave frequency.

The energy harvesters were fabricated simply by directly collecting PVDF nanofibers from the aluminum electrodes

placed on the slide glass, PET film, PEN film, and paper substrates. During the electrospinning process, the poling axes of the PVDF nanofibers are formed along each nanofiber axis by the combined action of mechanical stretching and electrical poling. Meanwhile, the mechanical vibration of the substrate resulting from the sound wave generates an alternating tension and compression along the axis of each PVDF nanofiber. The tensile stress produces tensile strain only along the nanofiber axis, but the compressive stress results in both compressive strain along the axis and bending. Thus, it can be imagined that the 33 piezoelectric mode (longitudinal mode) becomes excited upon application of both tension and compression to the PVDF nanofibers, while the 31 piezoelectric mode (transverse mode) becomes excited only upon application of compression.³⁹ The bending is negligible because the length of PVDF nanofibers is predominant over their diameter. In conclusion, t[he](#page-7-0) 33 piezoelectric mode is excited by the sound-driven mechanical vibration, resulting in a sinusoidal output voltage. The output voltage profiles in Figure $3(a)$ show the effects of the various substrates on the energy harvesters. The maximum output voltages shown in Figure [3](#page-3-0)(b) increased gradually from the slide-glass- to the paper-based harvester. It can be postulated that the more a substrate [is](#page-3-0) deformed the higher its output voltage. At a glance, substrate material stiffness affects deformation most significantly. Intrinsic substrate stiffness is related mainly to the Young's modulus, and extrinsic stiffness is a function of substrate thickness. As an intrinsic parameter, Young's modulus is a measure of the stiffness of a material. Note that Young's moduli for the substrate materials used in previous works are as follows: glass 73 GPa (by Paul et al.⁴⁰), PET 3.5 GPa (by Karger-Kocsis et al.⁴¹), PEN 5-5.5 GPa (by Sidler et al.⁴²), paper 5.2−7.1 GPa (by Yokoyama et al.⁴³[\). A](#page-7-0) material with a lower Young's m[odu](#page-7-0)lus will show larger deformatio[n u](#page-7-0)nder the same load. By extension, a device [ba](#page-7-0)sed on a substrate with a lower Young's modulus will likely exhibit a higher sound-driven maximum voltage. The result in Figure $3(c)$ is, however, contradictory to this expectation. Rather, the substrate with the higher modulus showed a higher maximum [vo](#page-3-0)ltage, with the exception of the glass substrate. This may also be caused by differences in substrate thickness. Thus, we investigated the maximum voltages of the PET film $(100 \ \mu m)$ and PEN (105 μ m) to assess the intrinsic effect more precisely (shown below). Interestingly, the cycling output voltage curves and the maximum voltages of the PET (100 μ m) and PEN (105 μ m) films were almost identical even though the Young's modulus of the PEN was 43−57% higher than that of the PET. (Note that the maximum voltages of the PEN $(100 \mu m)$ and PEN (105 μ m) were 0.3463 and 0.3498 V, respectively.) Thus, the effect of Young's modulus appears to be less important. Note that the piezoelectric performance on the styrene− butadiene−styrene (SBS) rubber substrate, which has a Young's modulus in the MPa range, was not assessed because the substrate did not vibrate in a manner synchronized with the sound wave. Thus, the substrate material requires a certain amount of elasticity for synchronization with sound waves. The nonlinear relationship between substrate thickness and output voltage (Figure $3(e)$), on the other hand, is conspicuous and consistent with conventional expectations. Thinner substrates, regardless of the[ir](#page-3-0) intrinsic stiffness, can output higher voltages, so paper (which is abundant and which showed good piezoelectric performance (Figure 3)^{44,45}) is the most suitable substrate for fabrication of high-efficiency nanogenerators.

Figure 5. (a) Schematic of the paper-based nanogenerator and (b−d) the sound-wave-induced mechanical deformation profiles of (b) 76, (c) 66, and (d) 33 μ m thick paper substrates.

We fabricated 76, 66, and 33 μ m thick paper-based energy harvesters to evaluate the effects of paper thickness on their piezoelectricity. As shown in Figure $4(a)$, the output voltage amplitudes increased gradually with decreasing paper thickness. To investigate the trend and limit of t[he](#page-4-0) substrate thickness, we evaluated piezoelectric performance using a 9 μ m thick PET film substrate and compared the results against those of substrates of 9, 100, and 255 μ m thicknesses. The output waveforms shown in Figure 4(b) are sinusoidal in shape, and the maximum voltage shown in Figure $4(c)$ increases dramatically as the substrate [th](#page-4-0)ickness decreases. On the basis of the results for the 9 μ m thick PET film, [w](#page-4-0)e expect the piezoelectric performance of PVDF nanofibers on paper thinner than 33 μ m to be superior to those on a thicker

substrate. As shown in Figure $4(c)$, the energy harvesters fabricated with the 76, 66, and 33 μ m thick papers output maximum voltages of 0.3167[,](#page-4-0) 0.4885, and 0.7781 V, respectively. We used a vibration model to numerically simulate the paper substrates to understand how the energy harvesters deformed in response to the sound waves.²⁹ The schematic in Figure $5(a)$ shows the paper-based device structure (i.e., the PVDF filament(s) laid across the alumin[um](#page-7-0) electrodes). The detailed simulation parameters are provided in the Experimental Section. The paper-substrate harmonic deformation was relatively inhibited at a paper thickness of 76 μ m. [Namely,](#page-1-0) [one small \(<50](#page-1-0) nm high) hill formed along the width and up toward the center, while two <150 nm deep tapered valleys formed in a downward direction. The theoretical amount of

stress applied to the PVDF filament was calculated to be 34 Pa. The 66 μ m thick paper substrate harmonically deformed to a slightly greater extent: one small valley at the center and two small hills (one on each side) formed. The theoretical amount of stress applied to the filament also increased slightly to 35.3 Pa. The deformation of the 33 μ m thick paper, however, was more complex and pronounced. The hills and valleys alternated widthwise and lengthwise, and the theoretical amount of stress applied to the filament improved significantly to 40.3 Pa. Consequently, paper mechanical deformation, which is a function of paper thickness, directly affects the amount of stress applied to PVDF nanofibers: thinner papers result in higher applied stresses, which produce higher output voltages.

4. CONCLUSION

PVDF-based piezoelectric energy harvesters were prepared using an electrospinning process. Electrospun PVDF nanofibers showed a high β -phase fraction due to application of a high voltage during electrospinning. The effects of various substrates and their thicknesses on the piezoelectricity of sound-driven energy generators were examined systematically using materials of various thicknesses. The thinnest $(33 \mu m)$ thick) paper showed the best piezoelectric performance. The numerical analyses also strongly supported the effectiveness of use of the thinnest paper substrate to fabricate energy harvesters.

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Notes

The authors declare no competing financial interest.

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